

The Effects of Manganese Dopant Content and Calcination Temperature on Properties of Titania-Zirconia Composite

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Abstract: The effects of dopant content and calcination temperature on Mn-doped TiO₂-ZrO₂ structure and properties were successfully investigated. Composite of Mn-doped titania-zirconia was synthesized by sol-gel method. Titanium(IV) isopropoxide was used as the precursor of TiO₂, while zirconia powder was used as another semiconductor. MnCl₂·4H₂O was used as the source of dopant in this study. Various amounts of manganese were incorporated into TiO₂-ZrO₂ and calcination was performed at temperatures of 500, 700 and 900 °C. Synthesized composites were characterized by Fourier-transform infrared spectroscopy (FTIR), specular reflectance UV-Vis spectroscopy (SR UV-Vis), X-ray diffraction method (XRD) and scanning electron microscopy equipped with X-ray energy dispersive spectroscopy (SEM-EDX). The results showed that Mn-doped TiO₂-ZrO₂ with the lowest bandgap (2.78 eV) was achieved with 5% of Mn dopant and calcined at 900 °C, while Mn-doped TiO₂-ZrO₂ with the highest bandgap (3.12 eV) was achieved with 1% of Mn dopant content calcined at 500 °C.

Keywords: Mn-doped TiO₂-ZrO₂; bandgap; manganese; ZrO₂; TiO₂

■ INTRODUCTION

Titanium dioxide (TiO₂) is a semiconductor that has been widely used as a photocatalyst material because of its interesting properties such as low cost, strong oxidizing power, non-toxicity, and chemical inertness [1]. Another photocatalyst material that has attracted attention is zirconium oxide (ZrO₂) which has good stability and easily produces holes in the valence band, leading to strong interactions with the active component. ZrO₂ is resistant to both alkali and acid conditions as well as oxidation and reduction reactions. Moreover, because of its properties, composites containing ZrO₂ as the main material have pulled much attention in recent years. This is mainly because of the difference in energy levels between ZrO₂ and other semiconductors to efficiently separate the charges, thus decrease the recombination of electrons and holes [2-3].

Previous research reported that coupling ZrO₂ with TiO₂ at Ti:Zr molar ratio of 1:1 had resulted in zirconium titanate hollow sphere with a non-visible-light response (3.31 eV), however, this material has a higher redox potential [1]. Pirzada et al. [2] successfully synthesized nanocomposite TiO₂/ZrO₂ with the lowest bandgap (3.26 eV, non-visible light response) with an average particle diameter particle of 10.5 nm yielding a higher redox capability. However, the photocatalytic performance of semiconductor can be improved with an approach to broaden the photo-response by doping with various metals [3]. Doping of semiconductor material with transition metals leads to reduction of bandgap and it will provide a good photocatalytic activity [4].

The metallic dopants with 3d orbitals, such as Mn, Fe, Co and Zn, had been studied and reported in recent years [5-9]. Transition metal will reduce the particle size

of photocatalyst and increase its surface area, Table 1 shows the use of dopants with 3d orbitals on the photocatalyst materials that had been studied previously. Doping photocatalyst with transition metals increases the surface defects and optical absorption of light, which can ultimately lead to an increase in the efficiency of the photocatalyst [9]. The lowest bandgap of TiO₂-ZrO₂ doped using Fe (2.83 eV), Co (2.94 eV) and Zn (2.87 eV) were achieved in composites with 7% of iron content calcined at 500 °C, 3% of cobalt content calcined at 500 °C, and 5% of zinc content calcined at 900 °C, respectively [6-8]. The properties of semiconductor are enhanced after manganese doping because Mn²⁺ and Mn³⁺ exist on the surface of photocatalyst and catch the electron in semiconductor. Therefore, Mn should trap electrons and suppress electron-hole recombination. Thus, the lower the electron-hole recombination is, the higher the quantum efficiency of the photons will be and then will reduce bandgap value [10]. Mragui et al. [5] successfully synthesized Mn-TiO₂ with 10% of Mn content and bandgap of 2.95 eV, using MnCl₂·4H₂O as a dopant source.

Beside the addition of dopant, the calcination temperature can also affect the bandgap value of the photocatalyst material by changing the structure of the material. TiO₂ which has been synthesized at high calcination (400 °C) could have higher photoactivity compared to the TiO₂ which was synthesized at low temperature (200 °C) [11]. However, photocatalytic activity is related to the combination properties of crystalline phase, particle size, surface area, electron-hole recombinant, bandgap energy, pore volume, and surface chemical state [11-12].

This study was conducted with various Mn dopant contents and calcination temperatures to determine its effects on the properties of Mn-doped TiO₂-ZrO₂ composite. The addition of Mn dopant and increased

calcination temperature were expected to produce a TiO₂-ZrO₂ composite with reduced bandgap energy. Mn-doped TiO₂ was developed on the surface of ZrO₂ by the sol-gel process. The sol-gel method was chosen because it has several advantages, such as an easy process, simple equipment, homogeneous phase, the ability to control the crystal size, and low temperature condition [13-14].

■ EXPERIMENTAL SECTION

Materials

Titanium(IV) isopropoxide (TTIP, 97%, Sigma-Aldrich) was used as TiO₂ precursor. Manganese(II) chloride tetrahydrate (MnCl₂·4H₂O, 98%, Merck) was used as source of Mn dopant. Zirconia powder (ZrO₂, Jiaozuo Huasu) was used as the source of coupling semiconductor. Absolute ethanol (Merck) and demineralized water (Jaya Sentosa) were used as solvents.

Instrumentation

Specular reflectance UV-Vis spectrophotometer (SR UV-Vis) UV1700 Pharmaspec was used to measure the ultraviolet and visible (UV-Vis) absorption of the material. Fourier-transform infrared spectroscopy (FTIR) analysis was conducted using Thermo Nicolet iS10. The crystal structures of material were analyzed by using X-ray powder diffractometer (XRD) PANalytical X'Pert PRO MRD instrument with Cu K α radiation. Surface morphology was observed with scanning electron microscopy equipped with energy dispersive X-ray spectrometer (SEM-EDX) Phenom Desktop ProXL.

Procedure

Mn-doped TiO₂-ZrO₂ was synthesized by the sol-gel method. Firstly, 2.5 mL of TTIP was dissolved in 25 mL of absolute ethanol and then stirred homogeneously. One gram of ZrO₂ powder was dispersed into 25 mL of demineralized water then as much as 14.38, 43.59, 72.66, 101.72, and 130.79 mg of MnCl₂·4H₂O was separately added dropwise into the mixture to obtain 1, 3, 5, 7 and 9% Mn wt./Ti wt., respectively. The mixture was stirred for 30 min and separated by centrifugation at 2000 rpm for 1 h. The obtained solid was aged in the open space

Table 1. Bandgap data of Mn-TiO₂, Fe, Co and Zn doped TiO₂-ZrO₂ at the optimum condition

Sample	Eg (eV)
Mn-TiO ₂ 10% 500 °C	2.95 [5]
Co-ZrO ₂ -TiO ₂ 3% 500 °C	2.94 [6]
Fe-ZrO ₂ -TiO ₂ 7% 500 °C	2.83 [7]
Zn-ZrO ₂ -TiO ₂ 5% 900 °C	2.87 [8]

for 24 h and then heated at 60 °C for 24 h. Composites with various Mn dopant contents were calcined at 500 °C for 4 h to observe the effect of Mn contents. Additionally, the composite with 5% of Mn content was calcined at 700 and 900 °C to observe the effect of calcination temperature. The variation of dopant percentage and calcination temperature were based on the previous works [6-8]. All samples were characterized by using XRD, FTIR, SR UV-Vis and SEM-EDX.

■ RESULTS AND DISCUSSION

Fig. 1 displays the morphology and its corresponding EDX spectra of 5% Mn-doped $\text{TiO}_2\text{-ZrO}_2$ calcined at 500 °C. The morphology of Mn-doped $\text{TiO}_2\text{-ZrO}_2$ appears to be rough and its particles appear to be agglomerated with an estimated diameter of 1–5 μm .

Table 2 shows the result of EDX spectra analysis. EDX analysis confirms that $\text{TiO}_2\text{-ZrO}_2$ was successfully synthesized and 0.33% of Mn was present in the composite.

FTIR spectra of Mn-doped $\text{TiO}_2\text{-ZrO}_2$ with various contents of Mn dopant and calcination temperatures are shown together in Fig. 2. FTIR spectra show hydroxyl stretching vibration of H_2O around 3300–3700 cm^{-1} and hydroxyl bending vibration of H_2O around 1650 cm^{-1} [15]. The Zr–O vibration was found around 420 and 744 cm^{-1} correspond to monoclinic structure of ZrO_2 [3]. At low wavenumber region, the absorption belonged to the vibration of Ti–O–Ti, Mn–O–Ti and Mn–O are found around 400–700 cm^{-1} [16]. Spectra of Mn-doped $\text{TiO}_2\text{-ZrO}_2$ has combined characteristic vibrations between TiO_2 and ZrO_2 , where there are broad stretching vibration

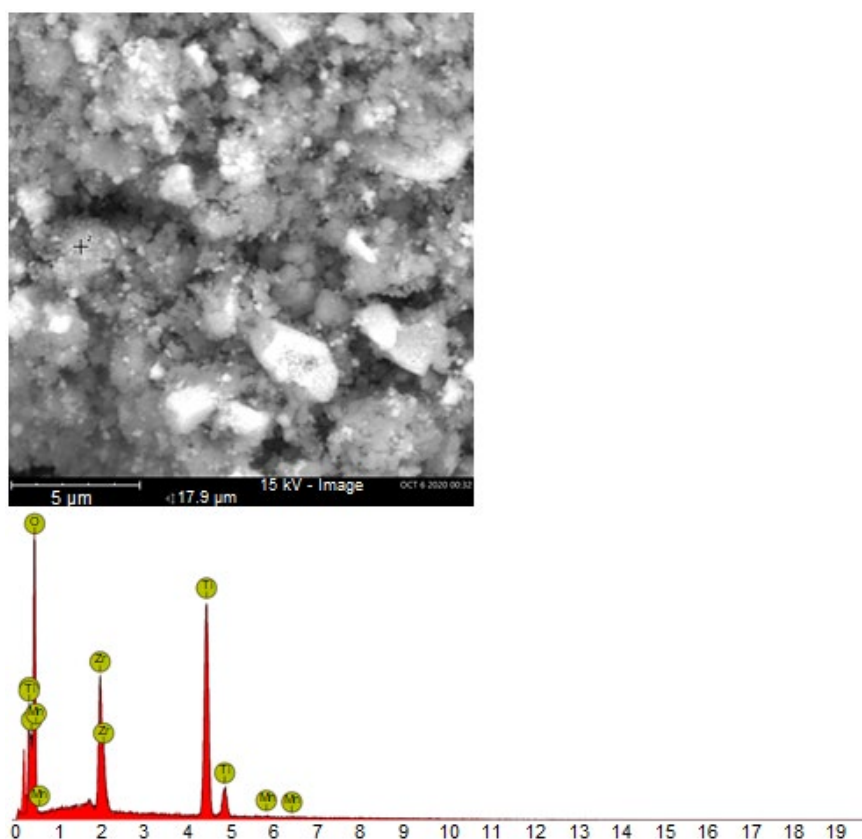


Fig 1. SEM image of $\text{TiO}_2\text{-ZrO}_2$ with 5% Mn content calcined at 500 °C and its corresponding EDX spectra

Table 2. EDX analysis of 5% Mn-doped $\text{TiO}_2\text{-ZrO}_2$ calcined at 500 °C

Sample	% Weight				
	Zr	Ti	O	Mn	Total
5% Mn-doped $\text{TiO}_2\text{-ZrO}_2$ @ 500 °C	49.09	32.42	18.16	0.33	100

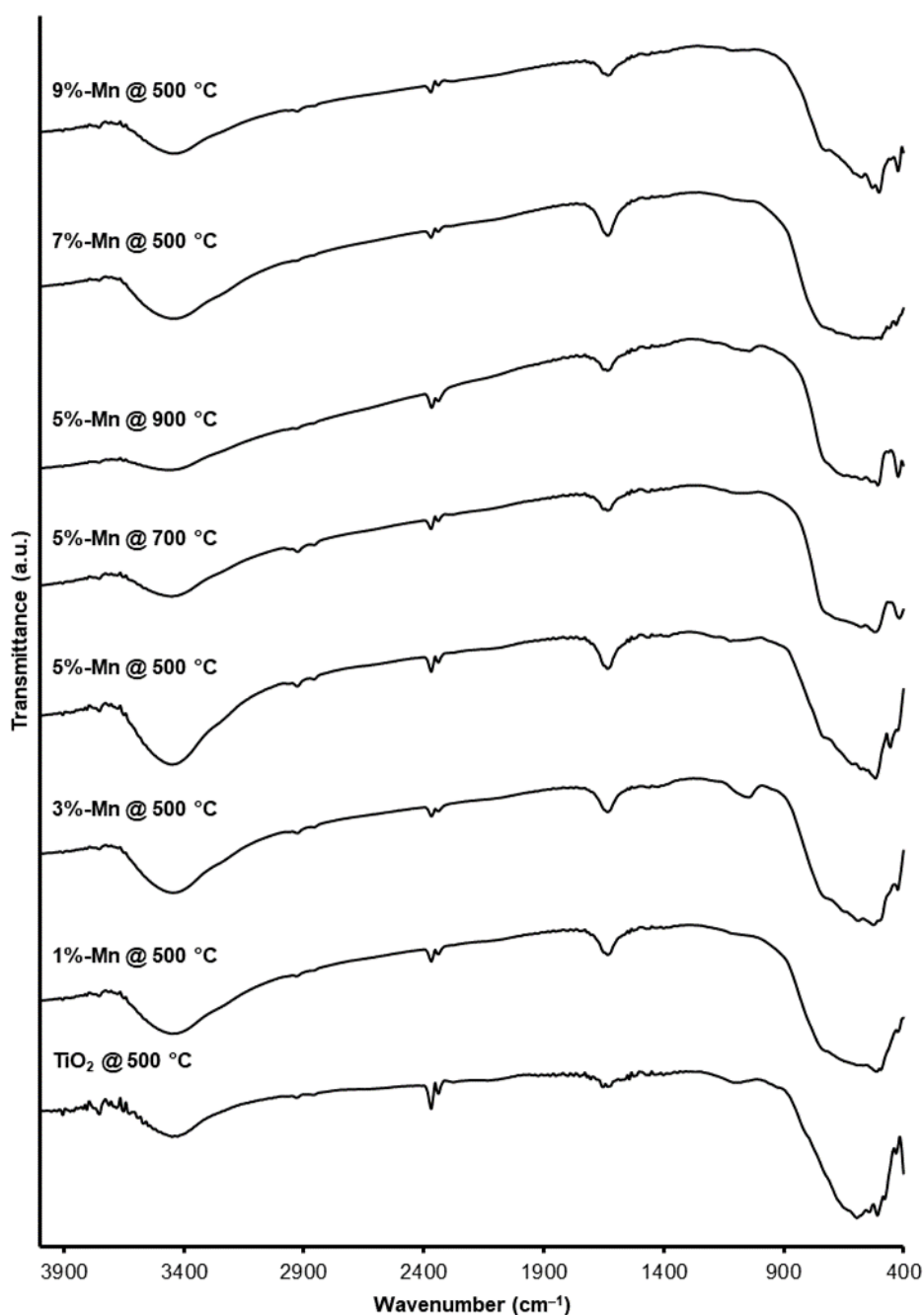


Fig 2. FTIR spectra of Mn-doped $\text{TiO}_2\text{-ZrO}_2$ with various dopant contents and calcination temperatures

of OH at $3300\text{--}3700\text{ cm}^{-1}$ and also around $400\text{--}700\text{ cm}^{-1}$, which is probably the vibration of Zr-O-Zr, Ti-O-Ti, or Mn-O-Ti. The stretching and bending intensities of H_2O decrease at 9% of Mn-doped $\text{TiO}_2\text{-ZrO}_2$, possibly because the number of Mn replacing Ti increases. The hydroxyl bending and stretching intensities of H_2O at 1630 and $3300\text{--}3700\text{ cm}^{-1}$, respectively, decrease with the increase of calcination temperature. This possibly happened

because H_2O molecules of synthesized material are released due to the high temperature heating during the calcination process.

Fig. 3 shows diffraction patterns of various Mn-doped $\text{TiO}_2\text{-ZrO}_2$ composites, while Table 3 presents their calculated crystallite sizes based on Scherrer's equation [17]. The diffraction patterns confirm the presence of monoclinic ZrO_2 and anatase TiO_2 . There are

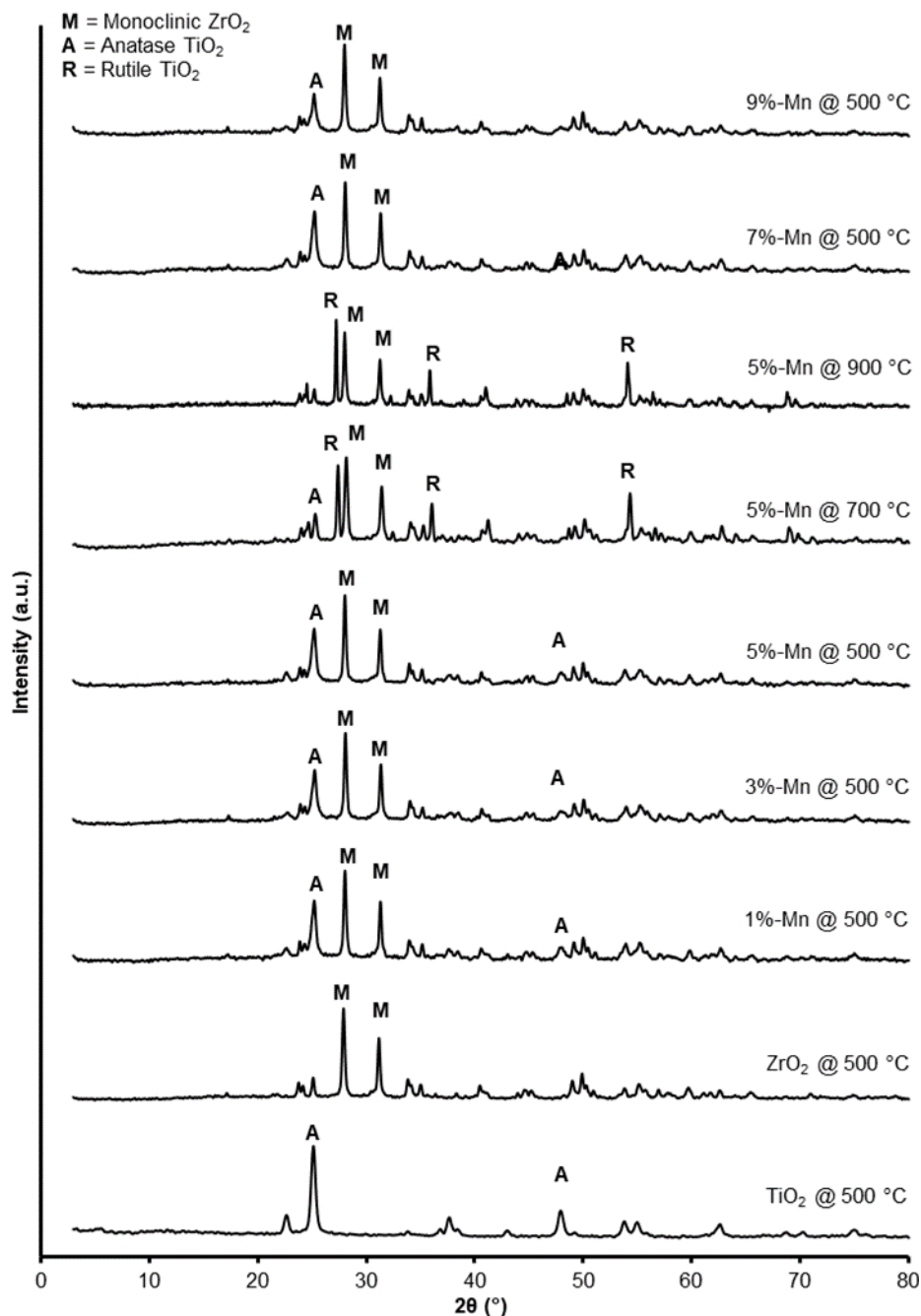


Fig 3. Diffraction patterns of Mn-doped $\text{TiO}_2\text{-ZrO}_2$ with various dopant contents and calcination temperatures

no tetragonal and rutile patterns observed at 500 °C. Monoclinic peaks are observed at 28.21° (-111) and 31.50° (111), according to ICDD:01-083-0944, while anatase peaks are observed at 25.28° (101) and 48.05° (200), according to ICDD:00-021-1272. All Mn-doped $\text{TiO}_2\text{-ZrO}_2$ had a pattern combination of high monoclinic and anatase intensities at 500 °C, while no Mn pattern was observed. The high intensity of monoclinic peaks shows

that ZrO_2 has a greater amount than anatase. The average crystallite sizes of monoclinic and anatase remain unchanged significantly as the content of manganese dopant increases. It indicates that Mn^{2+} (67 pm) has substituted Ti^{4+} (60.5 pm) in the anatase TiO_2 structure, as the ionic radius of both metal ions are similar [18]. Some literature reported that TiO_2 starts to form rutile structure at 500 °C [7,11]. One report found

Table 3. Average crystallite size of various Mn-doped TiO₂-ZrO₂

Sample	Crystal phase	(hkl)	L (nm)
ZrO ₂ @ 500 °C	Monoclinic	(-111)	26.36
TiO ₂ @ 500 °C	Anatase	(101)	15.72
1%-Mn @ 500 °C	Monoclinic	(-111)	25.10
	Anatase	(101)	15.00
3%-Mn @ 500 °C	Monoclinic	(-111)	27.94
	Anatase	(101)	14.96
5%-Mn @ 500 °C	Monoclinic	(-111)	28.82
	Anatase	(101)	15.60
5%-Mn @ 700 °C	Monoclinic	(-111)	22.88
	Anatase	(101)	18.44
	Rutile	(110)	29.19
5%-Mn @ 900 °C	Monoclinic	(-111)	30.62
	Rutile	(110)	44.64
7%-Mn @ 500 °C	Monoclinic	(-111)	28.28
	Anatase	(101)	15.72
9%-Mn @ 500 °C	Monoclinic	(-111)	29.01
	Anatase	(101)	15.12

that the presence of Mn²⁺ ion promotes anatase-rutile phase transformation [16], while another report suggested that Zr⁴⁺ prevents phase transformation of anatase [19]. The presence of ZrO₂ is assumed to be responsible for preventing the anatase-to-rutile transformation on Mn-doped TiO₂-ZrO₂ calcined at 500 °C.

The diffraction patterns also confirm the presence of ZrO₂ monoclinic, TiO₂ anatase and TiO₂ rutile at 5% Mn-doped TiO₂-ZrO₂ calcined at 700 and 900 °C. Monoclinic and anatase peaks were observed similar to the previous discussion, but the intensity of anatase decreased while the rutile increased. Rutile peaks are observed at 27.91° (110), 36.43° (101) and 55.11° (211), according to ICDD:01-088-1175 [20]. The intensity of anatase decreased as the calcination temperature increased because of the anatase-to-rutile transformation. The rutile pattern starts to be observed at 700 °C and almost all anatase peaks transform into rutile at 900 °C. The average crystallite size of monoclinic ZrO₂ also remains unchanged at higher calcination temperature. On the other hand, there is an increase in rutile crystallite size, which is possibly due to agglomeration of the rutile phase at 900 °C.

The SR UV-Vis absorption spectra of the Mn-doped TiO₂-ZrO₂ are shown in Fig. 4 with the variation of Mn

dopant content and calcination temperature, while Table 4 presents the bandgap energy values. The bandgap energy of the composites was evaluated by using absorption edge cross method. All synthesized materials have lower E_g values than TiO₂ (3.12 eV) and ZrO₂ (3.23 eV) [8,21]. The SR UV-Vis spectra show that all synthesized materials have absorption edge at visible range (above 400 nm). The presence of Mn²⁺ ions forms a new energy orbital under the conduction band and decreases the E_g value of the materials [22]. The increased calcination temperature reduces the E_g values of Mn-doped TiO₂-ZrO₂. This is possibly because the higher calcination temperature would facilitate the phase transformation of anatase to rutile and decrease the E_g value [23].

Based on the obtained results, the synthesized Mn-doped TiO₂-ZrO₂ using the sol-gel method had been successfully carried out. The Mn dopant was proven to reduce the E_g values from 3.12 to 2.78 eV with an optimum content of 5% and calcination temperature of 900 °C. The addition of Mn dopant 1–7% was shown to be able to reduce the E_g value, and it may possibly increase the response of the synthesized material under visible-light range. The E_g values of Mn-doped TiO₂-ZrO₂ increased as the addition of Mn dopant reached above 7%

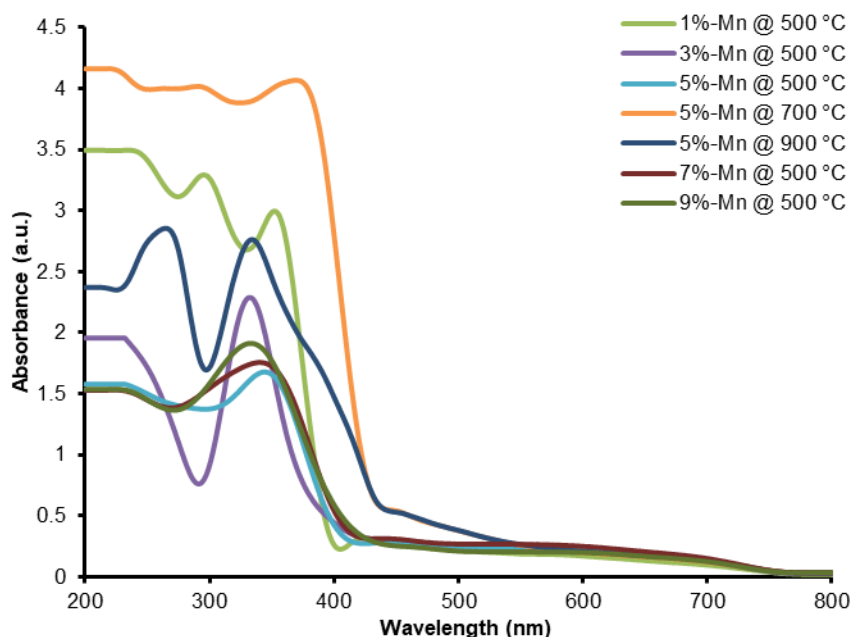


Fig 4. SR UV-Vis absorption spectra of various Mn-doped $\text{TiO}_2\text{-ZrO}_2$

Table 4. Bandgap energy data of Mn-doped $\text{TiO}_2\text{-ZrO}_2$

Sample	E_g (eV)
1%-Mn @ 500 °C	3.12
3%-Mn @ 500 °C	3.09
5%-Mn @ 500 °C	3.03
5%-Mn @ 700 °C	2.87
5%-Mn @ 900 °C	2.78
7%-Mn @ 500 °C	2.81
9%-Mn @ 500 °C	2.95

due to the heterojunction effect appearing and cancelling the dopant effect [4,6,21]. Furthermore, the higher calcination temperature was proven to enable the material transformations phase of anatase to rutile, which causes the E_g value to be smaller. This result corresponds with other previous works in which the addition of dopant between 1–9% and calcination temperature would show optimum condition of doped composite [4,6-8].

CONCLUSION

The investigation of Mn dopant content and calcination temperature affecting the properties of Mn-doped $\text{TiO}_2\text{-ZrO}_2$ composite had been successfully conducted. The results of EDX analysis confirm the presence of Zr, Ti, O and Mn. It was confirmed by the XRD analysis that the presence of Zr^{4+} inhibits the

anatase-to-rutile transformation at higher calcination temperature. The bandgap of Mn-doped $\text{TiO}_2\text{-ZrO}_2$ decreases as the calcination temperature and the dopant content increase until reaching the optimum condition. The composite with the lowest bandgap is Mn-doped $\text{TiO}_2\text{-ZrO}_2$ with 5% of Mn content calcined at 900 °C. All of the synthesized composites in this research can be considered as a potential visible-light-responsive photocatalyst.

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